

MECHANICAL PROPERTIES OF TRIAXIAL BRAIDED CARBON/EPOXY COMPOSITES

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ABSTRACT

In an on-going effort to increase the safety and efficiency of turbine engines, the National Aeronautics and Space Administration is exploring lightweight alternatives to the metal containment structures that currently encase commercial jet engines. Epoxy reinforced with braided carbon fibers is a candidate structural material which may be suitable for an engine case. This paper reports flat-coupon mechanical-property experiments performed to compliment previously reported subcomponent impact testing and analytical simulation of containment structures. Triaxial-braid T700/5208 epoxy and triaxial-braid T700/M36 toughened epoxy composites were evaluated. Also, two triaxial-braid architectures ($0^\circ \pm 60^\circ$, $0^\circ \pm 45^\circ$) with the M36 resin were evaluated through tension, compression, and shear testing. Tensile behavior was compared between standard straight-sided specimens (ASTM D3039) and bowtie specimens. Both double-notch shear (ASTM D3846) and Iosipescu (ASTM D5379) tests were performed as well. The M36/ $0^\circ \pm 45^\circ$ configuration yield the best response when measurements were made parallel to the axial tows. Conversely, the M36/ $0^\circ \pm 60^\circ$ configuration was best when measurements were made perpendicular to the axial tows. The results were used to identify critical properties and to augment the analysis of impact experiments.

KEY WORDS: Epoxy Resin, Carbon Fiber, Mechanical Properties

1.0 INTRODUCTION

The engine case is the largest structural component of a turbine engine. The case is critical for its roles in aerodynamic efficiency, structural integrity, and safety. Obviously the rotating components of a turbine engine are designed to minimize the likelihood of failure. However, the consequences of such a failure could be catastrophic and it is imperative that the engine case contain any potential fragments as well as withstand the significant overloads subsequent to a rotating component failure. A significant knowledge base supports the design of metal engine cases; the introduction of lighter, composite cases will require the development of experimental data and design expertise to compete with the established technology. Investigations at NASA

Glenn Research Center have delved into subcomponent testing methodology and material screening [1,2], developing design input [3], and dynamic structural analysis of case-impact events [4]. Reference [2] reported the results of high-strain energy impact experiments on braided $0^\circ/\pm 60^\circ$ carbon fibers/M36 toughened epoxy composites. Damage in the composite panels appeared to initiate as tensile-fiber failures on the back face of the panel at the lower-range impact velocities. The extent of the back-face fiber failure increased with increasing impact velocity until through thickness tears opened and allowed the projectile to penetrate the composite panel. Interestingly enough, there was no evidence of between-ply delamination in any of the panels and the through-thickness cracks did not extend much beyond the region of impact. The absence of delamination and extensive crack-growth encourage further investigation of this composite system.

Optimizing the composite constituents and fiber architectural arrangements through component or subcomponent testing alone would be a time consuming and costly endeavor. Finite element analyses have been employed to understand the salient features of the impact experiments [1] and to simulate the damage [5]. This experimental program was conceived to provide basic mechanical data on the composites for use in these analyses. Testing was performed on composite systems with two different resins and two different fiber architectures: $0^\circ/\pm 60^\circ$ /M36, $0^\circ/\pm 60^\circ$ /5208, and $0^\circ/\pm 45^\circ$ /M36. Specimens from two of the three composites systems were taken from panels which had been previously used for impact testing. Using the impact-tested panels ensured the identical material was being evaluated, but care had to be taken to obtain specimens without preexisting impact damage.

2.0 FABRICATION AND EXPERIMENTAL TECHNIQUES

Three carbon/epoxy composite systems were investigated. Braided $0^\circ/\pm 60^\circ$ and $0^\circ/\pm 45^\circ$ preforms were combined with M36 resin (toughened epoxy system from HEXCEL Composites). Also a Cytec 5208 resin (non-toughened epoxy system from CytecFiberite) reinforced with braided $0^\circ/\pm 60^\circ$ fibers was investigated. Composites were fabricated by resin infusion into six layers of a triaxial braided T700 carbon fiber preform. The preforms were manufactured by A&P Technology and the composites were fabricated by GE Aircraft Engines. The $0^\circ/\pm 60^\circ$ braided preform had 12k flat tow fibers in the $\pm 60^\circ$ (bias) directions and 24k flat tow fibers in the 0° (axial) direction. The $0^\circ/\pm 45^\circ$ braided preform had 12k flat tow fibers in the $\pm 45^\circ$ (bias) directions and 36k flat tow fibers in the 0° (axial) direction. Both lay-ups were designed to have a fiber-volume fraction of 56.5%. In the $0^\circ/\pm 60^\circ$ system, 33.3% of the fibers were in the -60° direction, 33.3% were in the $+60^\circ$ direction, and 33.3% were in the 0° direction. Since the fiber volume in each direction was equal, both the individual plies and the entire lay-up were quasi-isotropic. In the $0^\circ/\pm 45^\circ$ system, 48.5% of the fibers were in the $\pm 45^\circ$ direction and 51.5% were in the 0° direction. The spacing between axial tows was 0.89 cm (0.352 in) in both layups. The spacing between the perpendicular repeating unit was 0.52 cm (0.203 in) for the $\pm 60^\circ$ layup and 0.89 cm (0.352 in) for the $\pm 45^\circ$ layup. Composite panels were formed from six plies with the 0° fibers aligned. The cured composites had a nominal thickness of about 3.2 mm (0.125 in), though there were systematic thickness variations associated with the braid pattern. Composites were fabricated in the form of 61 cm x 61 cm (2 ft x 2 ft) flat plates. The panels will be referred to as 5208/ $\pm 60^\circ$, M36/ $\pm 60^\circ$, and M36/ $\pm 45^\circ$ for the remainder of the discussion.

The M36/ $\pm 60^\circ$ and M36/ $\pm 45^\circ$ composite panels were first used for impact testing as described in reference [2]. The panels were impacted and penetrated by a soft, gelatin projectile at 197.5 m/s and 263.3 m/s for M36/ $\pm 60^\circ$ and M36/ $\pm 45^\circ$ composite, respectively. Post-impact-testing ultrasonic inspection suggested that the impact damage was concentrated near the impact site and in regions which were visibly damaged. Mechanical test specimen locations were selected from undamaged regions of the panel, as illustrated in Figure 1. A panel not subjected to impact testing was used for the 5208/ $\pm 60^\circ$ specimens. The gage width of specimens was chosen based on repeating units of either the spacing between axial tows in the longitudinal direction or the tow width in the transverse direction. Straight-sided tensile specimens were 30.5 cm (12 in) long with a width of 2.54 cm or 4 repeating units, whichever was larger. Therefore, the transversely oriented M36/ $\pm 60^\circ$ had a width of 2.54 cm (1 in) and the width was 3.58 cm (1.41 in) for all other straight-sided tensile specimens. A bowtie specimen design was also used for tensile testing in an attempt to isolate cut fibers from the gage section. Figure 2 illustrates the shape and dimensions of the bowtie specimens; the gage width for all was 3 axial tow repeating units. The gage width in each of the bowtie specimens was chosen to contain three axial tows. The compression specimens were 2.54 cm wide and 7.62 cm long (1 in x 3 in). The Iosepescu specimens were 1.91 cm by 7.62 cm with a 1.14 gage section (0.75 in x 3 in x 0.45 in). The double notch shear specimens were 12.7 cm by 8.08 cm with a 0.635 cm unnotched ligament (0.5 in x 3.18 in x 0.25 in). Straight-sided tensile specimens were instrumented with a single axial and a single transverse strain gage. Compression specimens were instrumented with a single axial strain gage. Iosepescu specimens were instrumented with a single $\pm 45^\circ$ strain gage. The gage section of the bowtie specimens were painted with a grid of dots spaced 0.159 cm (0.0625 in) apart.

3.0 RESULTS AND DISCUSSION

The mechanical properties of the three composite panels were evaluated through tensile, compressive, and shear testing in both longitudinal and transverse directions by Cincinnati Testing Laboratory. Three specimens were tested for each condition. All tests were conducted in constant displacement-rate control.

3.1 Tensile and Compressive Behavior The results of the tensile, bowtie-tensile, and compressive tests were summarized in Table 1. The M36/ $\pm 60^\circ$ and M36/ $\pm 45^\circ$ tensile response curves were plotted in Figure 3. The M36/ $\pm 60^\circ$, M36/ $\pm 45^\circ$, and 5208/ $\pm 60^\circ$ compression stress-strain curves were plotted in Figure 4. Comparison of the straight-sided tensile response, Figure 3(a), and the straight-sided compressive response, Figure 4, suggested that the moduli are similar for all the composites with 0/ $\pm 60^\circ$ fiber reinforcement. Figure 3(a) and Figure 4 also showed the strength is ordered highest to lowest as M36 along the axial fibers, M36 perpendicular to the axial fibers, 5208 along the axial fibers, and 5208 perpendicular to the axial fibers. The same figures of straight-sided specimen behavior indicated that the M36/ $\pm 45^\circ$ specimens consistently yielded the highest strength and stiffness when tested parallel to the axial fibers and the lowest strength and stiffness when tested perpendicular to the axial fibers.

Table 1. Ultimate stress and strain, elastic modulus and Poisson's ratio for static tension and compression in the longitudinal (along 0° fiber) and transverse loading orientations. Values are averages of three test results.

Composite	Test Orientation	Ult. Stress MPa (ksi)	Failure Strain %	Modulus GPa (Msi)	Poisson's Ratio
M36 / ±60°	Tensile longitudinal	805 (116.7)	1.7	47 (6.86)	0.27
	Bowtie longitudinal	624 (90.56)	-	-	-
M36 / ±45°	Tensile longitudinal	895 (129.8)	1.3	68 (9.83)	0.62
	Bowtie longitudinal	976 (141.6)	-	-	-
M36 / ±60°	Comp. longitudinal	507 (73.46)	1.3	45 (6.53)	-
M36 / ±45°	Comp. longitudinal	591 (85.77)	1.1	61 (8.86)	-
5208 / ±60°	Comp. longitudinal	308 (44.70)	0.71	49 (7.05)	-
M36 / ±60°	Tensile transverse	456 (66.14)	1.2	44 (6.33)	0.29
	Bowtie transverse	800 (116.0)	-	-	-
M36 / ±45°	Tensile transverse	194 (28.16)	2.2	18 (2.61)	0.17
	Bowtie transverse	346 (50.24)	-	-	-
M36 / ±60°	Comp. transverse	422 (61.17)	0.99	46 (6.61)	-
M36 / ±45°	Comp. transverse	265 (38.5)	1.7	18 (2.62)	-
5208 / ±60°	Comp. transverse	229 (33.25)	0.6	41 (5.94)	-

The bowtie tensile tests, Figure 3(b), were included in this study because the optimum specimen design is not obvious for braided architectures. The cross-sections of the straight-sided specimens in this study contained a number of “cut” fibers. In the bowtie specimens, the grip tapered to the reduced section at an angle that complimented the off-axis fiber orientation. This led to a gage section with nearly all the fibers continuing to the gripped region. In each case, the bowtie specimens should yield a higher strength than the straight specimens. The bowtie strengths were higher for the longitudinally and transversely loaded M36/±45° and for the transversely loaded M36/±60°. However, the longitudinally loaded M36/±60° bowtie specimens had lower strengths than the straight-sided specimens. The effect of the bowtie specimen on strain response is more difficult to predict than the effect of strength; and more difficult to measure. The grids which were applied to the bowtie specimens had insufficient resolution to resolve any deformation. If the load frame is sufficiently stiff with respect to the test specimen, there should be a simple relationship between cross-head displacement and axial strain; however the simple relationship could not be applied to these specimens. Further analysis into the suitability of this bowtie-type design is required. Issues that require further scrutiny include stress field at the notch tip and the required width with respect to the unit cell size.

3.2 Shear Behavior The results of the double-notch shear and Iosepescu shear testing were summarized in Table 2. All of the in-plane (1-2 and 2-1) shear moduli were very consistent. The ultimate strength also agreed very well for the in-plane orientations of the isotropic M36/±60° composites. Likewise, the strength of the in-plane orientations of the isotropic 5208/±60° composites were just beyond the standard deviation of the three specimens. The deviation was larger for the non-isotropic 0° ±45° layup.

Table 2. Summary of shear responses where orientation 1 is parallel to the axial fibers, orientation 2 is perpendicular to the axial fibers in the plane of the panel, and orientation 3 is perpendicular to the axial fibers through the thickness of the panel.

Composite	Test Orientation	Ult. Stress MPa (ksi)	Ult. St. Dev. MPa (ksi)	Shear Modulus GPa (Msi)
M36 / $\pm 60^\circ$	Iosepescu 1-2	252 (36.59)	24 (3.44)	15 (2.22)
	Iosepescu 2-1	252 (36.58)	9.0 (1.3)	19 (2.72)
	DNS 1-3	54 (7.76)	6.8 (0.979)	-
M36 / $\pm 45^\circ$	Iosepescu 1-2	261 (37.8)	24 (3.43)	15 (2.18)
	Iosepescu 2-1	207 (30.06)	8.4 (1.22)	16 (2.33)
	DNS 1-3	40 (5.77)	7.5 (1.08)	-
5208 / $\pm 60^\circ$	Iosepescu 1-2	186 (26.95)	2.1 (0.31)	17 (2.41)
	Iosepescu 2-1	200 (28.98)	6.2 (0.90)	17 (2.43)
	DNS 1-3	33 (4.72)	6.7 (0.969)	-

4.0 SUMMARY

Tensile, compressive and shear properties were obtained for three composites and used to augment analysis of impact testing [5]. Care must be taken in applying these results since the M36/ $\pm 60^\circ$ and M36/ $\pm 45^\circ$ coupons came from panels after impact testing. On the basis of these results, the M36/ $\pm 45^\circ$ configuration had the best response for properties parallel to the axial tows. M36/ $\pm 60^\circ$ has the more isotropic behavior, as demonstrated in the Iosepescu shear data. Also the M36/ $\pm 60^\circ$ had higher strengths when tested transverse to the axial tows. All the mechanical properties were consistently lower for the untoughened resin (5208/ $\pm 60^\circ$) as compared to the toughened system (M36/ $\pm 60^\circ$). Future work will compare these coupon level test results to the subcomponent impact tests and the finite element analysis. The testing can screen candidate composites and suggest systems for engine cases, but the optimal fiber/resin system and fiber architecture will vary with engine case design. Therefore, the integration of experiments and analysis will be crucial for successful application of composite in an engine case.

5.0 REFERENCES

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6.0 FIGURES

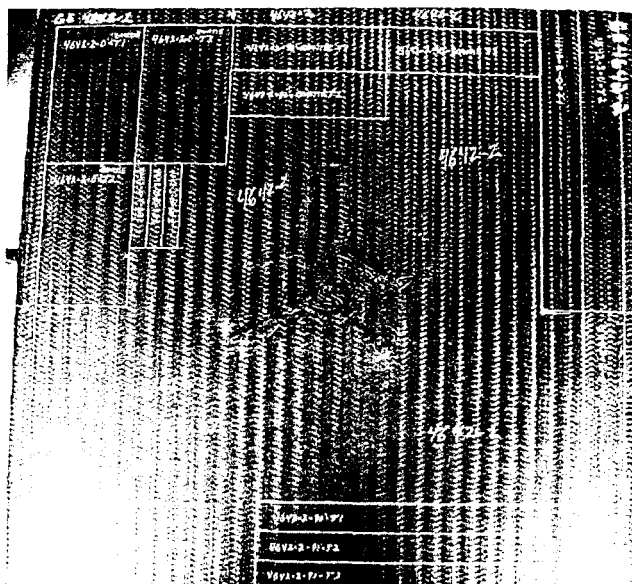


Figure 1. $\pm 60^\circ$ /M36 composite panel post impact with mechanical specimen locations designated.

Specimen	A	B	C	D	E
$\pm 60^\circ$ long-tensile	5.08 (2.0)	3.58 (1.411)	8.89 (3.5)	60°	2.68 (1.056)
$\pm 60^\circ$ trans-tensile	5.08 (2.0)	6.12 (2.409)	4.45 (1.75)	120°	1.55 (0.609)
$\pm 45^\circ$ long-tensile	5.08 (2.0)	6.21 (2.44)	8.89 (3.5)	90°	2.68 (1.056)

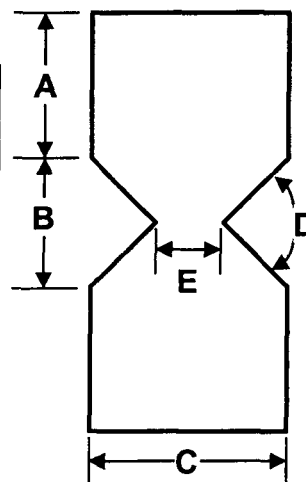


Figure 2. Specimen design and dimensions in centimeters (inches) for two longitudinal and one transverse bowtie configurations.

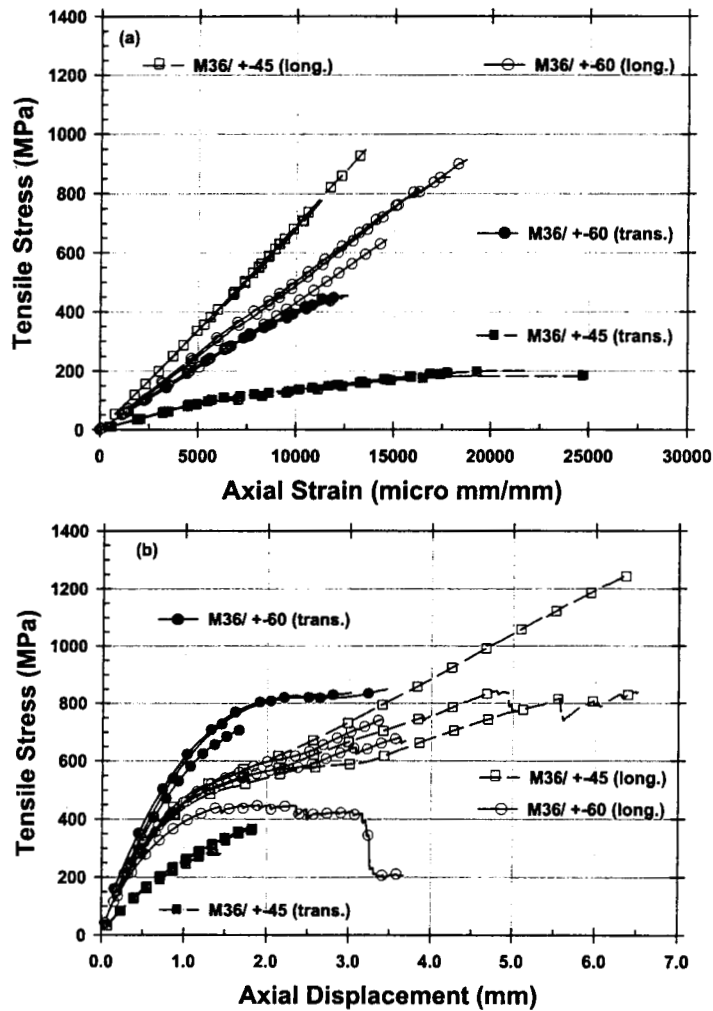


Figure 3. Results of (a) straight-sided and (b) bowtie tensile tests for M36 resin and both ± 60 and ± 45 fiber architectures. Precise strain not available with bowtie test configuration.

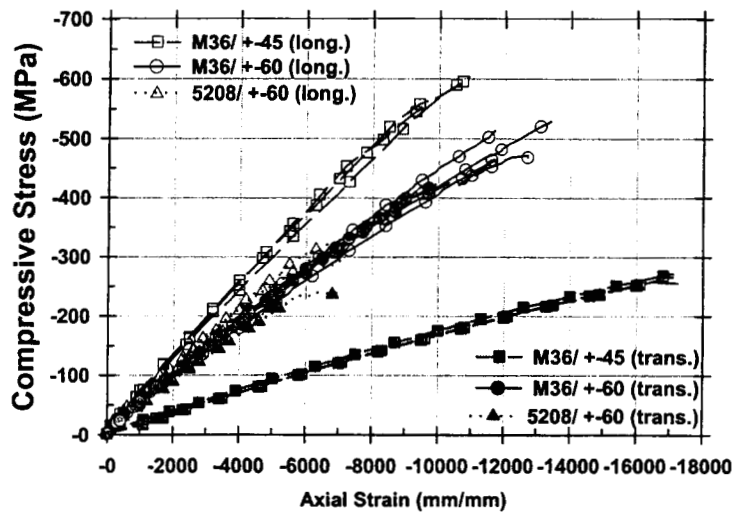


Figure 4. Compressive response of longitudinally and transversely oriented specimens.